

# Al-Si-Ti (Aluminum-Silicon-Titanium)

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The update of this ternary system by [2005Rag] presented a reaction sequence, a liquidus projection, a partial isothermal section at 1250 °C from the experimental results of [2004Bul], and six computed isothermal sections for Ti-rich alloys between 1200 and 700 °C from the thermodynamic description of [2002Aze]. Liquidus and solidus projections, a reaction sequence, and several isothermal sections were also presented in an updated review by [2006Per]. Recently, [2008Liu] reported new results on the liquidus projection and isothermal sections at 1000 and 900 °C.

## Binary Systems

The Al-Si system is of the simple eutectic type, with the eutectic temperature at 577 °C and 12.2 at.% Si. Recently, [2006Sch] carried out a new assessment of the Al-Ti system and presented a revised phase diagram. The intermediate phases in the system are:  $Ti_3Al$  ( $D0_{19}$ ,  $Ni_3Sn$ -type hexagonal, denoted as  $\alpha_2$ ),  $TiAl$  ( $L1_0$ , AuCu-type tetragonal, denoted  $\gamma$ ),  $TiAl_2$  (HfGa<sub>2</sub>-type tetragonal),  $TiAl_3$  (HT) ( $D0_{22}$ -type tetragonal), and  $TiAl_3$  (LT) (tetragonal, space group  $I4/mmm$ ). The Si-Ti phase diagram [Massalski2, 2008Liu] depicts the following compounds:  $Ti_3Si$  ( $Ti_3P$ -type tetragonal),  $Ti_5Si_3$  ( $D8_8$ ,  $Mn_5Si_3$ -type hexagonal),  $Ti_5Si_4$  ( $Zr_5Si_4$ -type tetragonal),  $TiSi$  ( $B27$ , FeB-type orthorhombic above  $\sim 800$  °C), and  $TiSi_2$  ( $C54$ ,  $TiSi_2$ -type orthorhombic).

## Ternary Phase Equilibria

[2003Gup] studied diffusion couples of pure Ti and an Al-Si eutectic alloy and identified the binary compounds and the ternary phases  $\tau_1$  and  $\tau_2$  of this system. With starting metals of 99.99% Al, 99.99% Si, and 99.98% Ti, [2008Liu] arc-melted more than 30 alloys under Ar atm. The alloys were annealed between 1000 and 550 °C for 2-3 weeks and quenched in water. The phase equilibria were studied with x-ray powder diffraction, energy dispersive x-ray analysis on a scanning electron microscope and differential thermal analysis (DTA) at a heating and cooling rate of 5 °C/min. The crystal structures, lattice parameters, and compositions of the identified phases were listed.

The isothermal sections constructed by [2008Liu] at 1000 and 900 °C are shown in Fig. 1 and 2. At 1000 °C (Fig. 1), the ternary phase  $\tau_2$  ( $ZrSi_2$ -type orthorhombic) is present. It forms incongruently at 1338 °C and has a composition range from  $Al_{21}Si_{46}Ti_{33}$  to  $Al_8Si_{59}Ti_{33}$  [2008Liu], with the lattice parameters of  $a = 0.35833$  nm,  $b = 1.3552$  nm, and  $c = 0.35793$  nm at  $Al_8Si_{59.6}Ti_{32.4}$ . At 900 °C (Fig. 2), the ternary phase  $\tau_1$  ( $Zr_3Al_4Si_5$ -type tetragonal; space group  $I4_1/amd$ ) is additionally present at the composition  $Al_8Si_{58.6}Ti_{33.4}$  with  $a = 0.35788$  and  $c = 2.7132$  nm. It melts incongruently between 953 and 1000 °C. The liquidus surface constructed by [2008Liu] from DTA data is shown in Fig. 3. The ternary phases  $\tau_2$  and  $\tau_1$  form through ternary peritectic reactions  $P_1$  and  $P_2$ ,

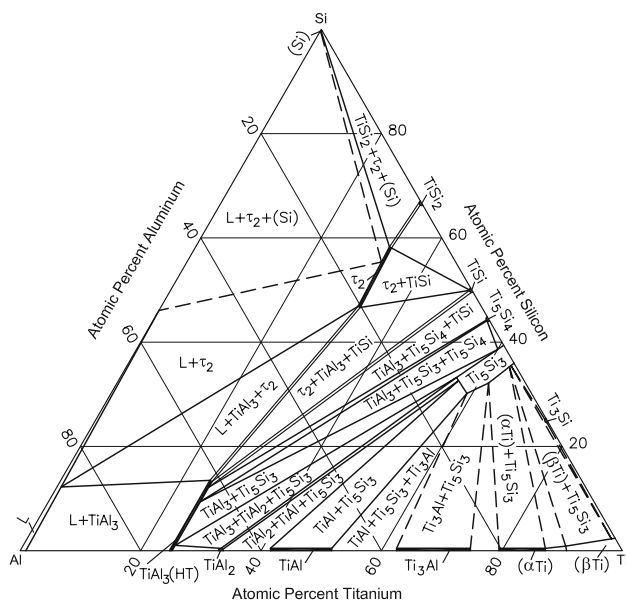


Fig. 1 Al-Si-Ti isothermal section at 1000 °C [2008Liu]

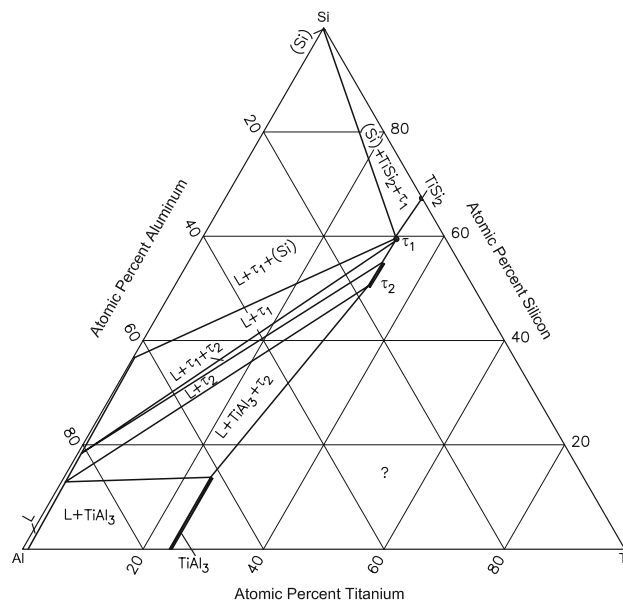
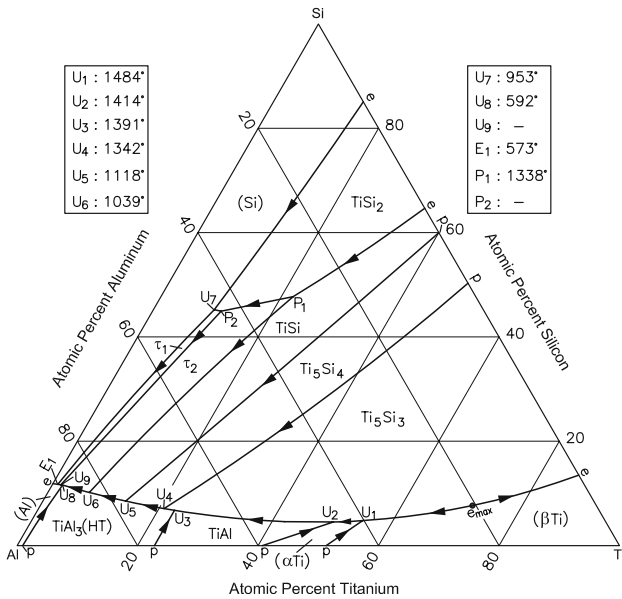


Fig. 2 Al-Si-Ti partial isothermal section at 900 °C [2008Liu]



**Fig. 3** Al-Si-Ti liquidus projection [2008Liu]

respectively. The incongruent formation of  $\tau_1$  through reaction  $P_2$ , however, could not be detected from DTA experiments. The solidification starts near the Ti end at a eutectic maximum  $e_{\max}$ :  $L \leftrightarrow (\beta\text{Ti}) + \text{Ti}_5\text{Si}_3$  at 1540 °C and

proceeds toward the Al corner through a series of transition reactions  $U_1$  to  $U_9$ . The final solidification is through the ternary eutectic reaction E:  $L \leftrightarrow (\text{Al}) + (\text{Si}) + \tau_1$  at 573 °C. A reaction sequence for the solidification range was given by [2008Liu].

## References

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